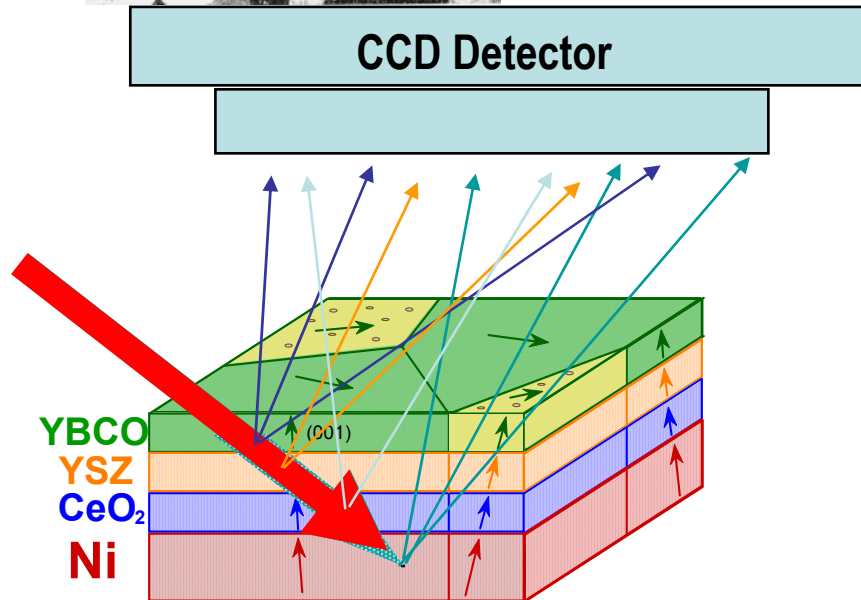
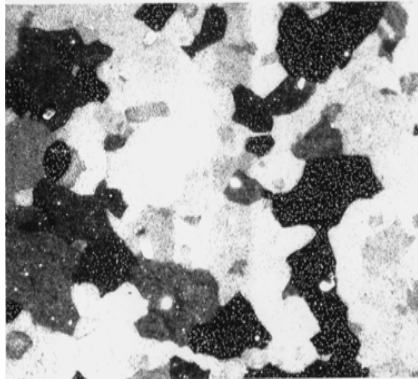
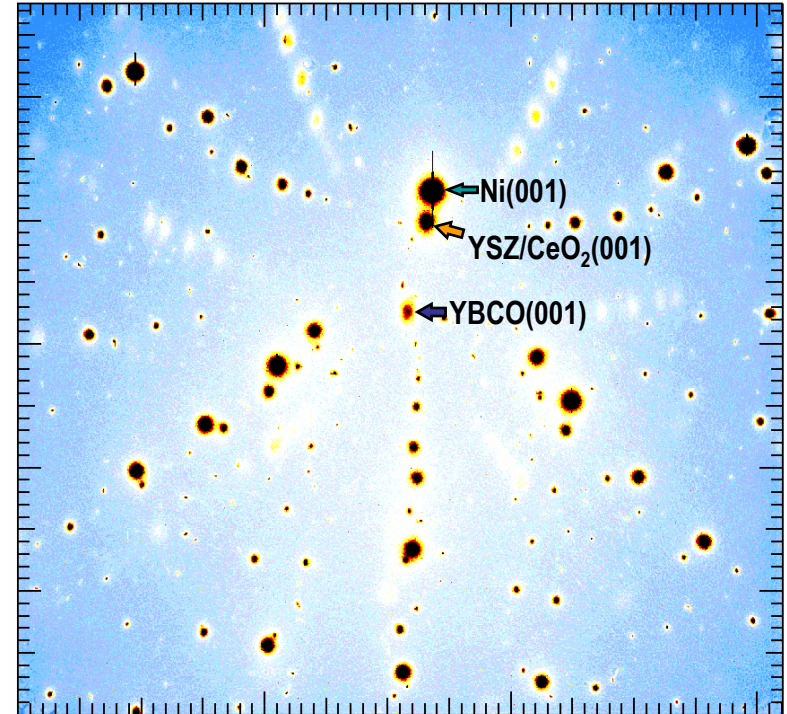


# Budai et al. polychromatic microdiffraction to epitaxial growth RABiTS

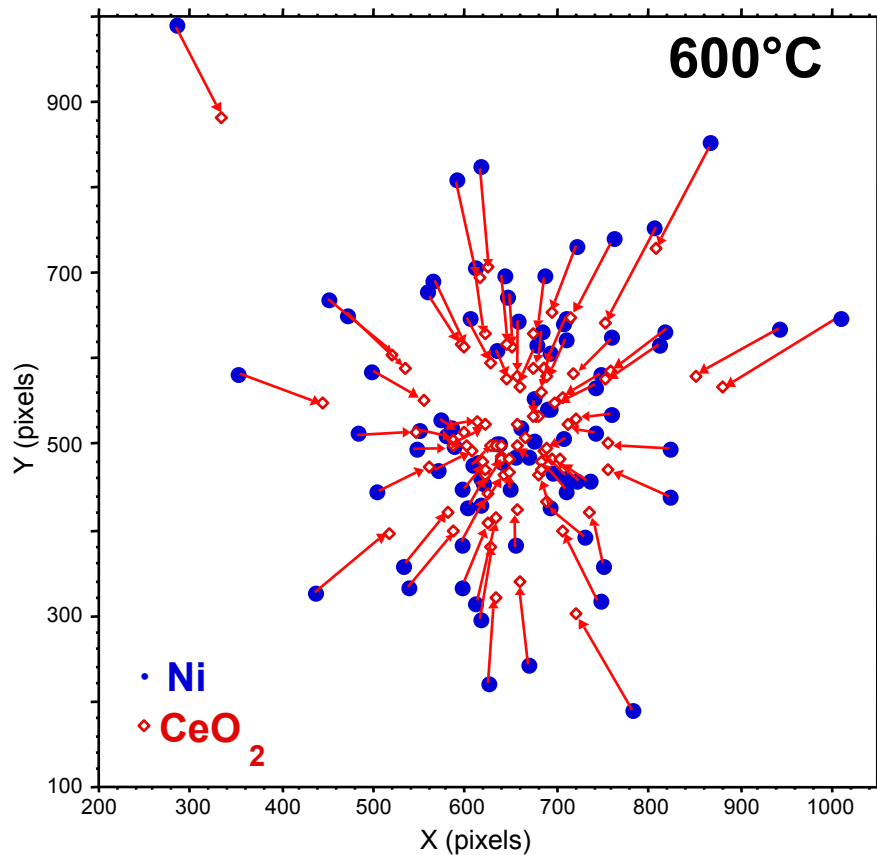
Optical:  $\sim 50\mu\text{m}$  grains



## CCD Laue Patterns

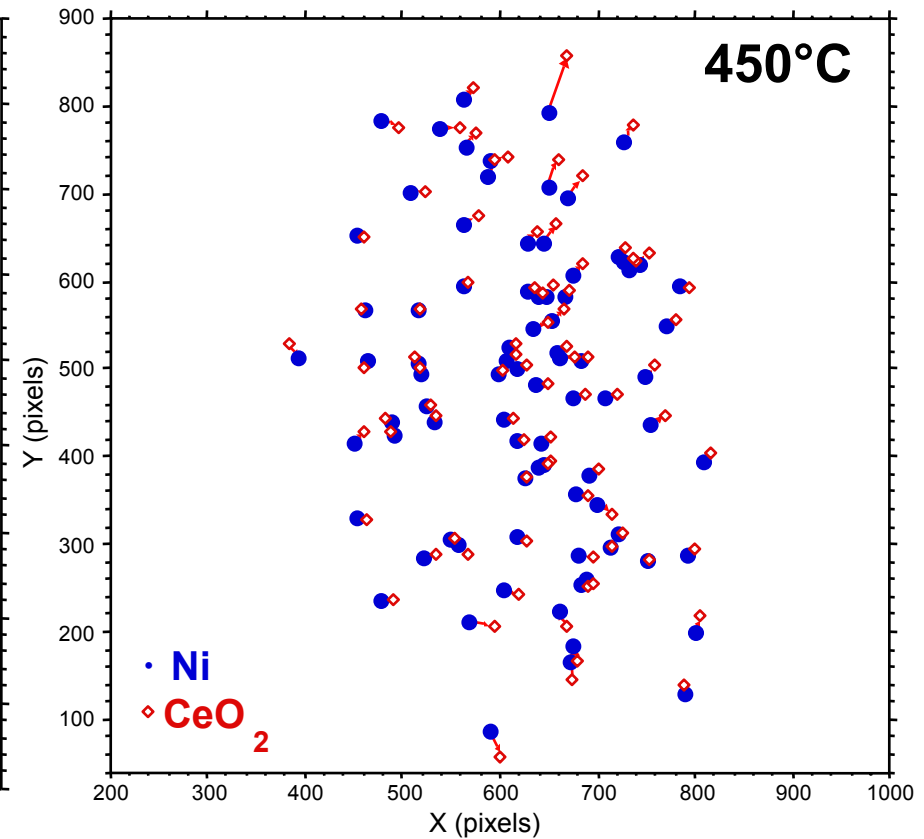


# Relative $\text{CeO}_2$ orientation depends deposition temperature



## High temperature growth:

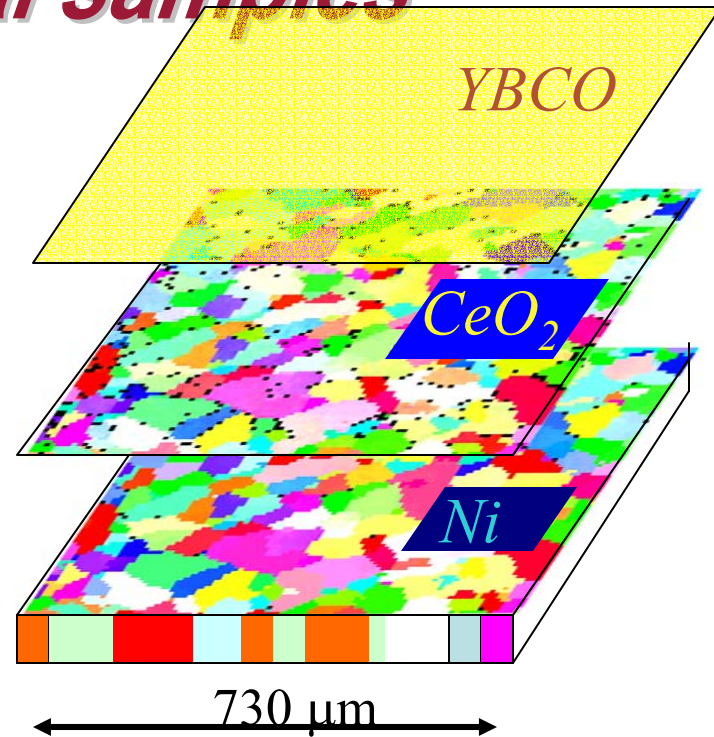
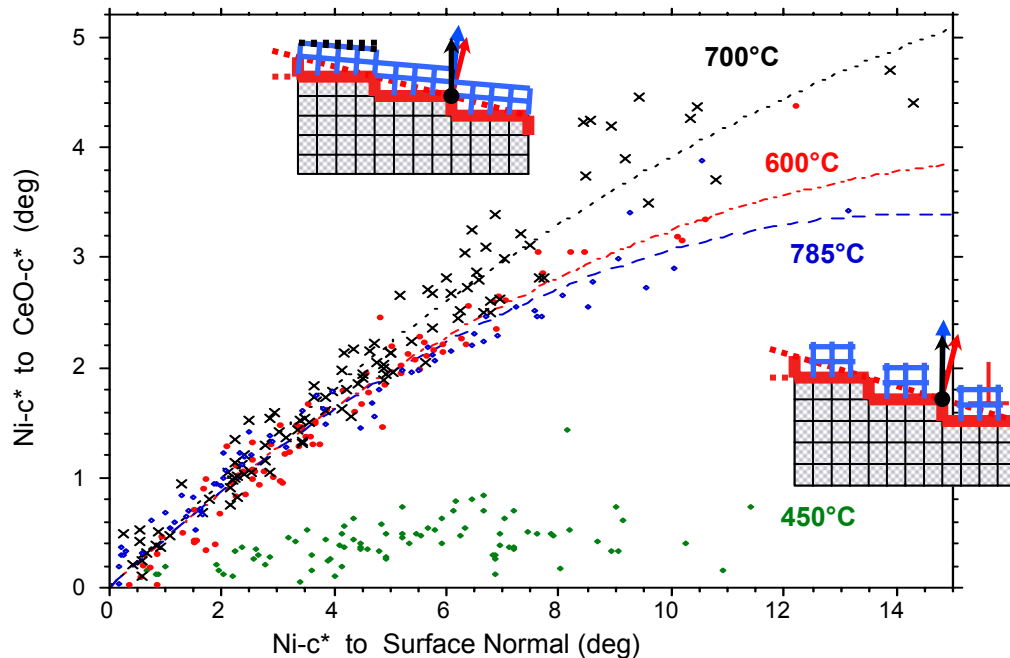
Crystallographic tilt towards  $\perp$   
Tilt increases monotonically with miscut



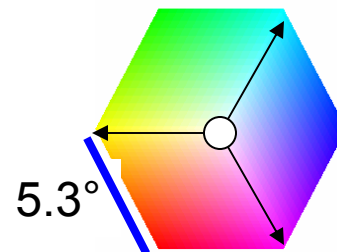
## Low temperature growth:

Small, ~biased tilts

# *Microbeam enables combinatorial measurements on real samples*

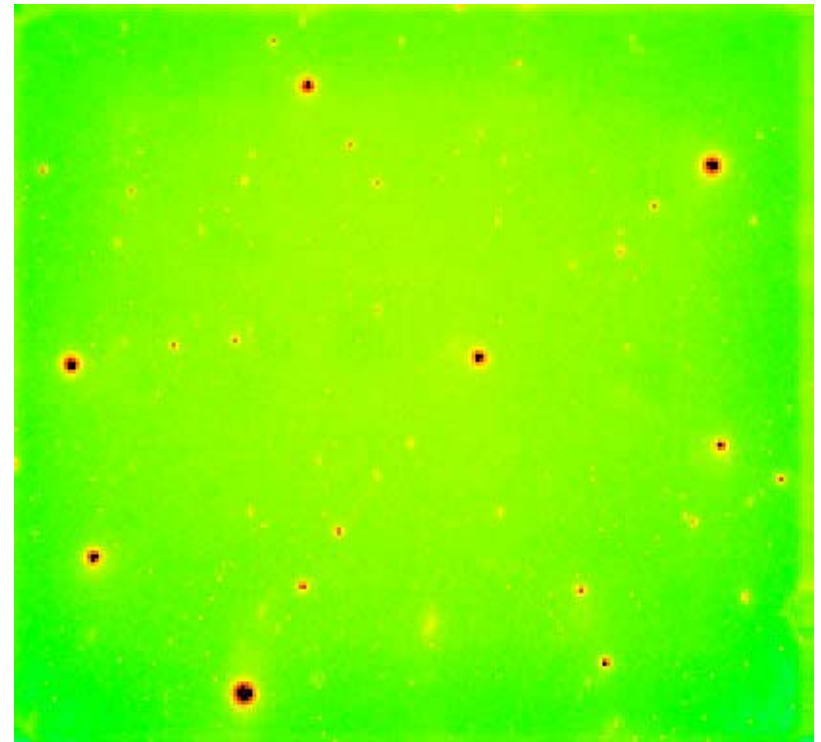
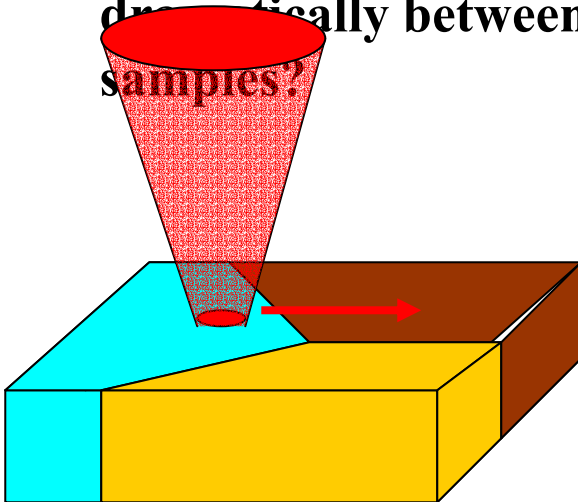


**Budai JD**, Yang WG, Tamura N, Chung JS, Tischler JZ, Larson BC, Ice GE, Park C, Norton DP **NATURE MATERIALS** 2 (7): 487-492 JUL 2003



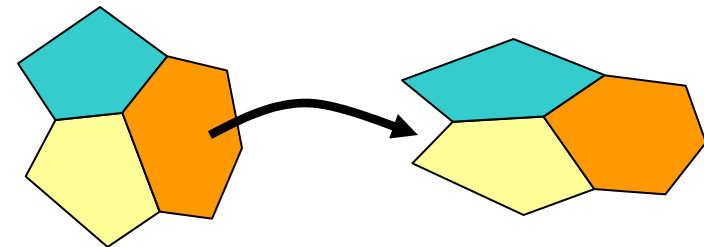
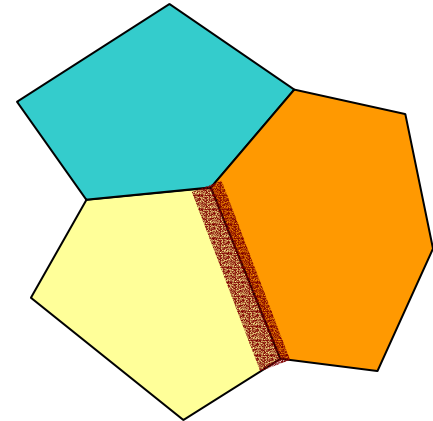
# Important questions remain

- Why does  $J_c$  decrease for thick samples?
- Why does mosaic on single Ni substrate grain differ dramatically between samples?



# How grain boundary/polycrystal networks interact - a major materials opportunity 21st century

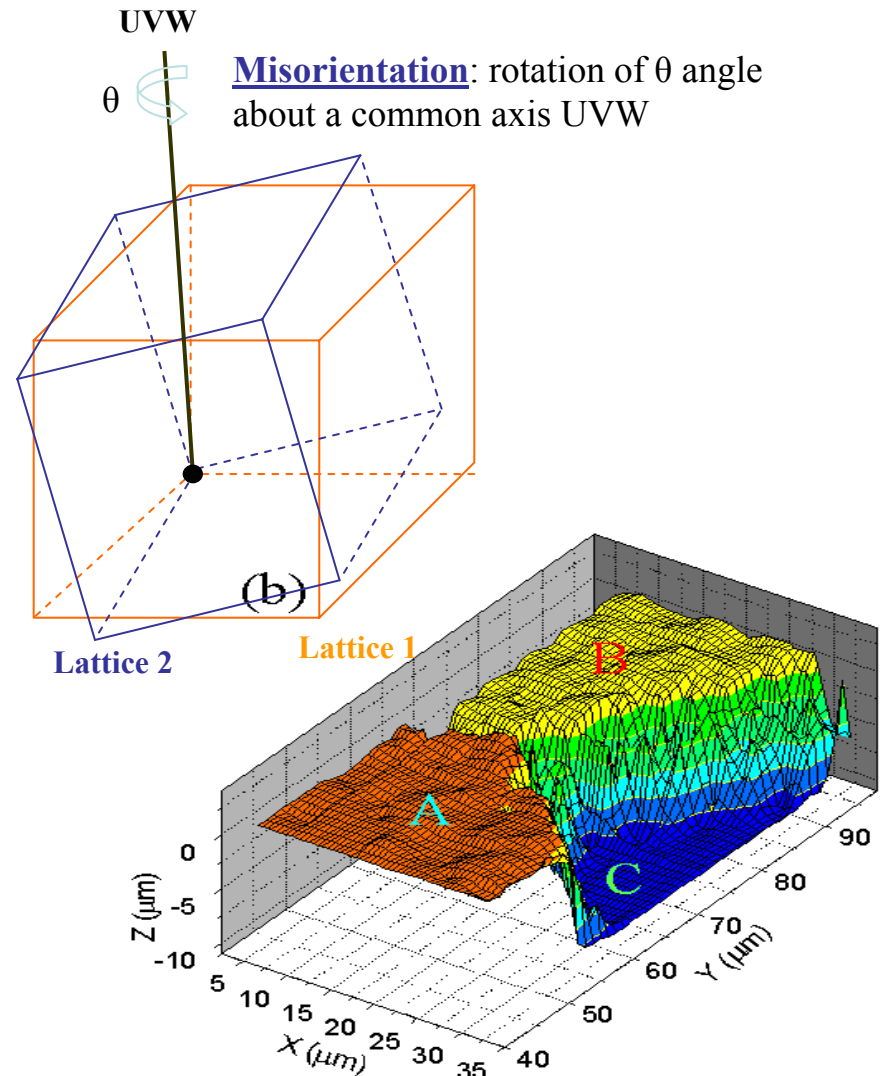
- What are the constitutive equations at grain boundaries?
  - How do they change with boundary type
- What are ideal microstructures?
  - How do different networks evolve during processing and in service?
- How can grain boundary distributions be controlled?
  - Grain boundary engineering



***Essential for nanophase and advanced layered materials***

# Unprecedented precision addresses long-standing issues/ tests CSL models

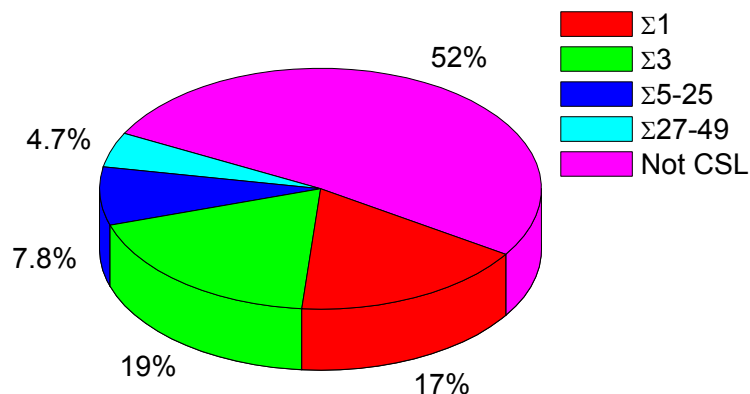
- CSL low-energy boundaries share lattice sites
  - $\Sigma$  denotes inverse fraction of shared sites
  - Theory: misorientation increases as  $\Sigma$  decreases
- Measured misorientation increase with  $\Sigma$
- Grain boundary normals
  - Ideal directions should have lower energy
  - Faceting may remove energy advantage



Morphology of Ni triple junction

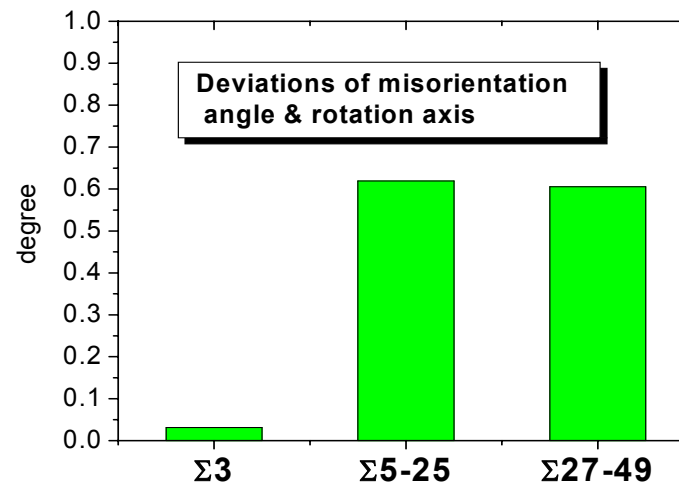


# Significant statistical information emerging



Total: 70

About 50% are CSLs, and 20% are found to be tilt, twist or having low-index in both grains.



No	Sigma type	Rotation Angle (degree)	Rotation angle off (degree)	Rotation Axis (RAX)	Rotation axis off (degree)	Boundary Normal (BN) in bi-crystal	Angle between RAX – BN (degree)	
B2	Σ21b	44.40	0.01	2, 1, 1	2.95	1.00, 0.32, 0.30 / 0.69, 1.00, 0.17	86.3	Tilt
B6	Σ47b	43.66	0.80	3, 2, 0	6.11	1.00, 0.07, 0.53 / 1.00, 0.87, 0.31	74.6	Tilt
B10	Σ37c	50.57	0.14	1, 1, 1	4.55	0.08, 1.00, 0.26 / 1.00, 0.12, 0.68	57.6	
B34	Σ1	(6.16°)				0.00, 1.00, 0.17 / 0.04, 0.27, 1.00	88.3	Tilt
A57	Σ3	60.00	0.01	1, 1, 1	0.02	1.00, 0.11, 0.02 / 0.32, 1.00, 0.87	86.4	Tilt
A314	Σ3	60.00	0.01	1, 1, 1	0.03	0.28, 0.31, 1.00 / 0.36, 0.37, 1.00	2.4	Twist

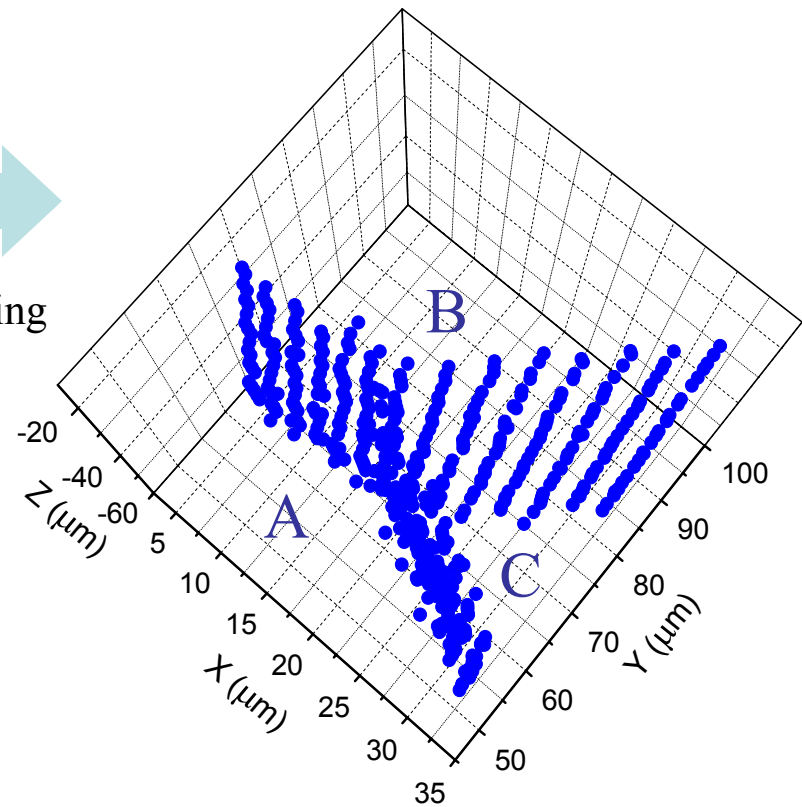
## Open questions:

1. Why and how are the deviations from ideal CSL model as  $\Sigma$  type increases?
2. Are there residual strains imposed near the deviated CSL boundaries?
3. Any difference of CSLs between near or below sample surface?
4. ....

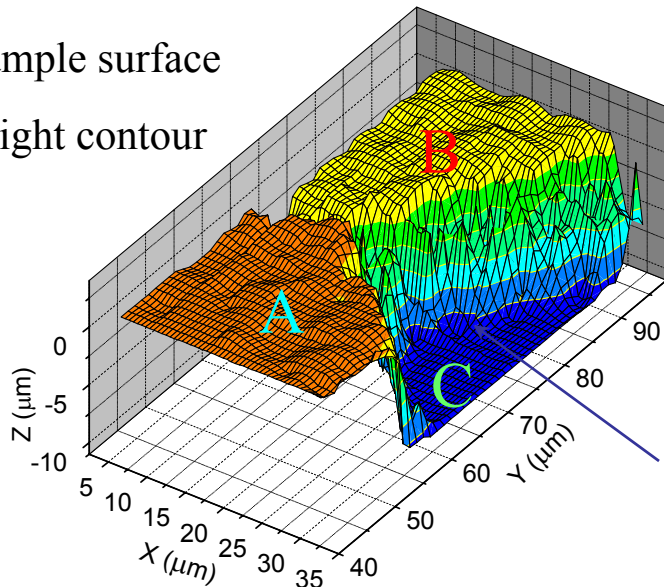
# Three Dimensional Morphology of Triple Junction



3-D mapping



Sample surface  
height contour



surface step  
between  
two grains

Misorientation angles:

A-B:  $16.572^\circ$

B-C:  $12.907^\circ$

C-A:  $5.538^\circ$



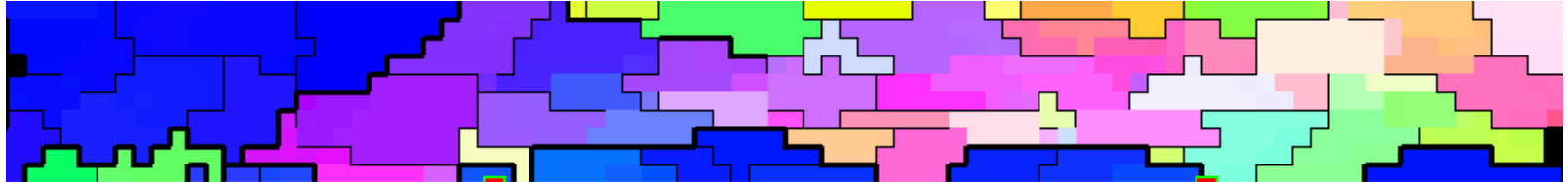
Polycrystalline grain structure can now be measured  
nondestructively in 3D-submicron resolution-meso scale

QuickTime™ and a  
Video decompressor  
are needed to see this picture.

# Thermal Grain Growth in Hot-Rolled Aluminum

1  $\mu\text{m}$  pixels, Boundaries: 5° & 20°

Anneal 250°C, 1 hr



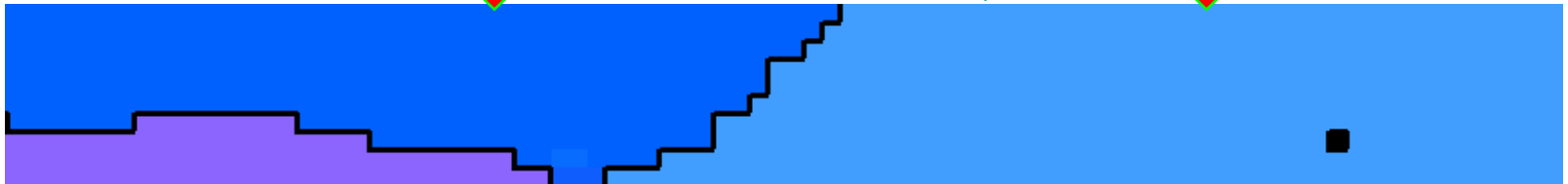
Anneal 350°C, 1 hr



Anneal 355°C, 1 hr



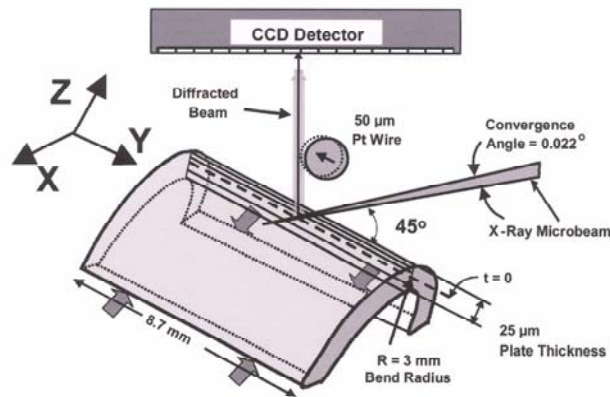
Anneal 360°C, 1 hr



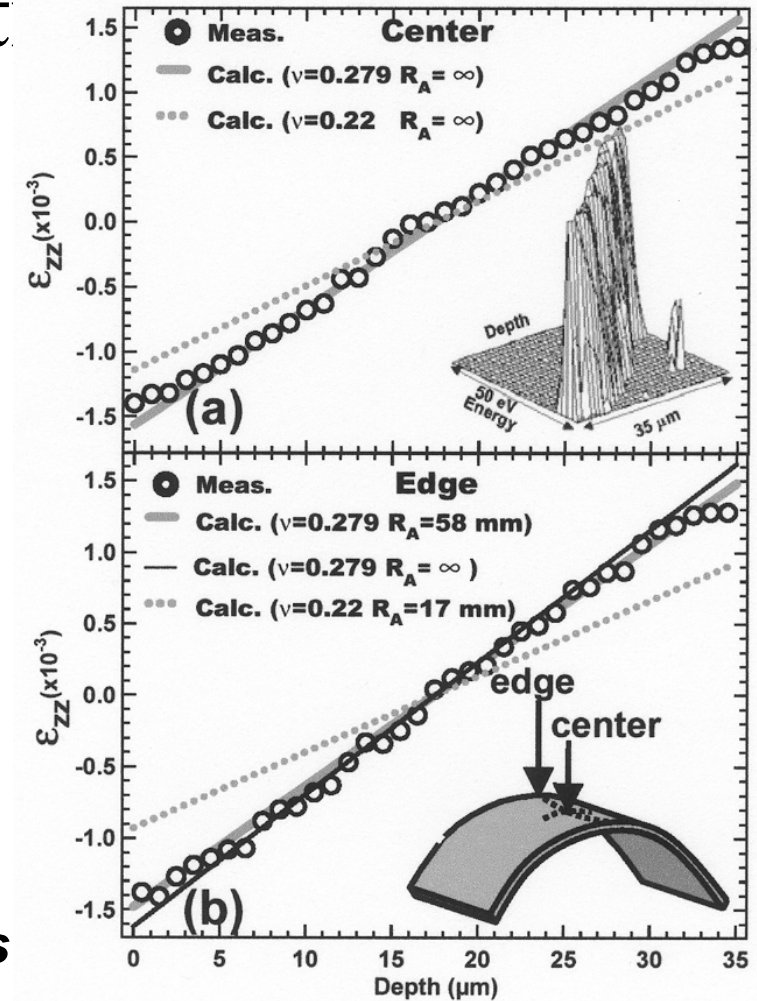
- GB motions include both high-angle and low-angle boundaries
- Complete and detailed 3D evolution needed for validation of theories.

# Elastic strain key driving force- Monochromatic DAXM measures intra- granular elast

- Local strain-even in single crystal
- Ultra-high precision local orientations
- Independent of grain orientation
- Phase sensitive

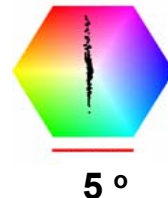
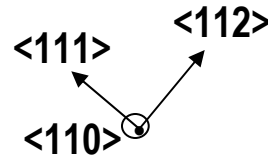
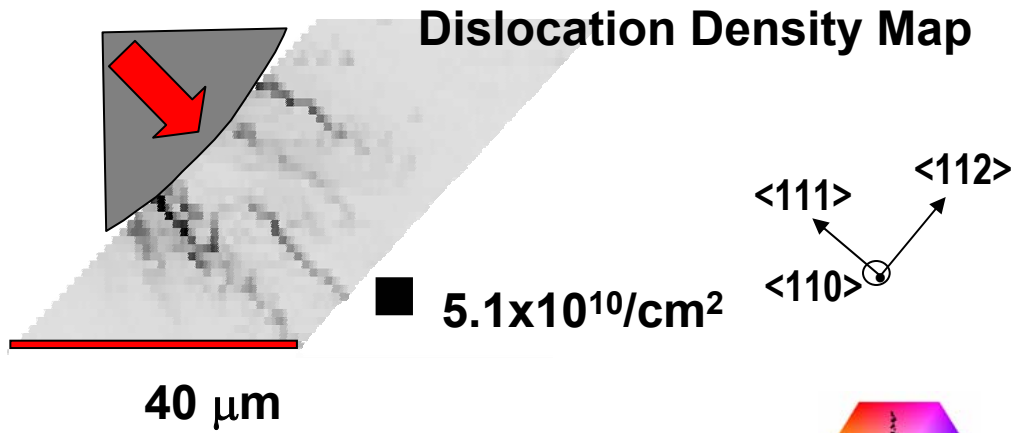
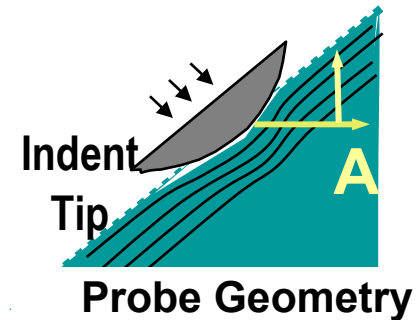


***Revolutionizes ability to study materials***

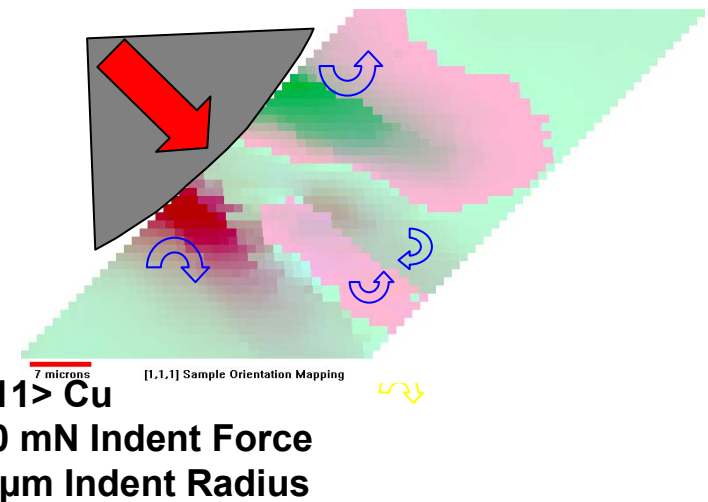


# Nanoindent in single crystals provides major insights into 3D deformation/modeling

- Deformation boundary conditions completely known/ volume modelable
- Best models predict some features not others-highly reproducible
- Single, bi-crystal, or polycrystal
- Strain-gradient models directly testable



## Lattice Rotation Map

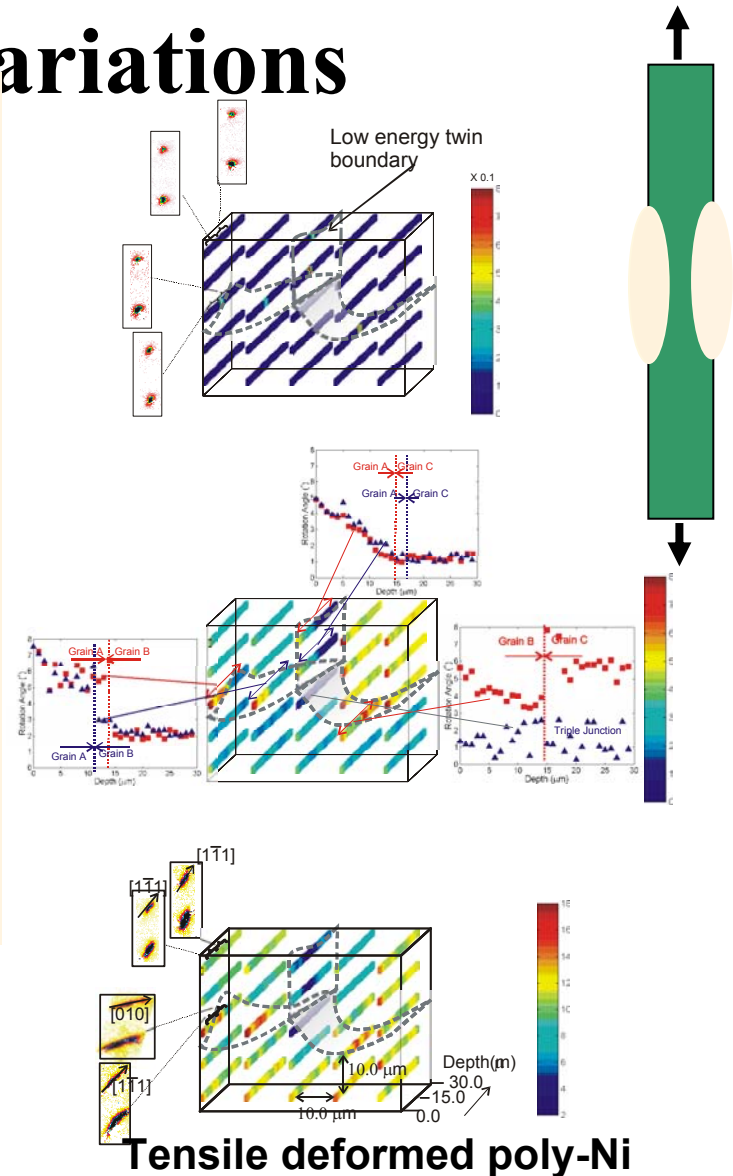


# intra-granular variations

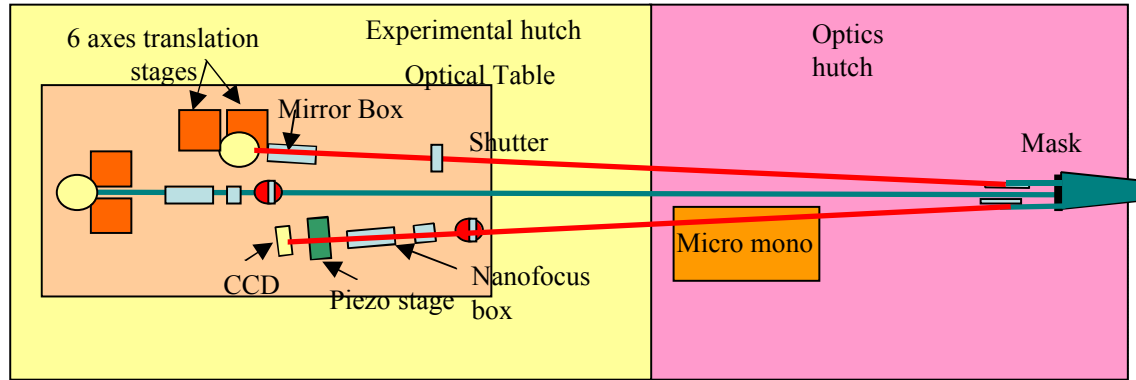
- Dramatic changes in deformations within single grain
  - Consistently large rotations near surface
- Plastic *and* elastic deformation measured
  - Essential information for understanding mechanisms
- Extensive sample characterization required for full boundary conditions

## Proposed research

- Full boundary conditions
- Low deformation
- Integrate theory



# To achieve potential and meet emerging demand - new microbeam lines and hardware proposed



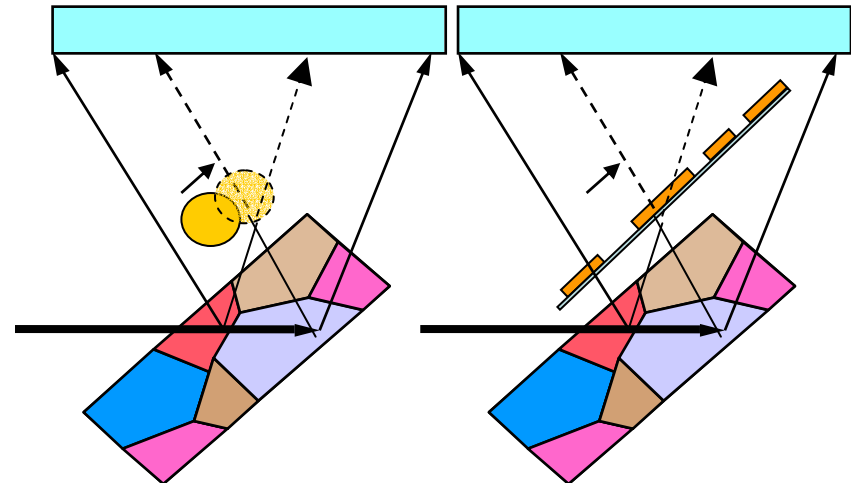
**Multiplexed 3D polychromatic diffraction-center for mesoscale research-**

- BM
- Operated by APS
- Greater general user access

**Spatial resolution 50nm→10nm**

**Accelerated 3D characterization 100-1000x**

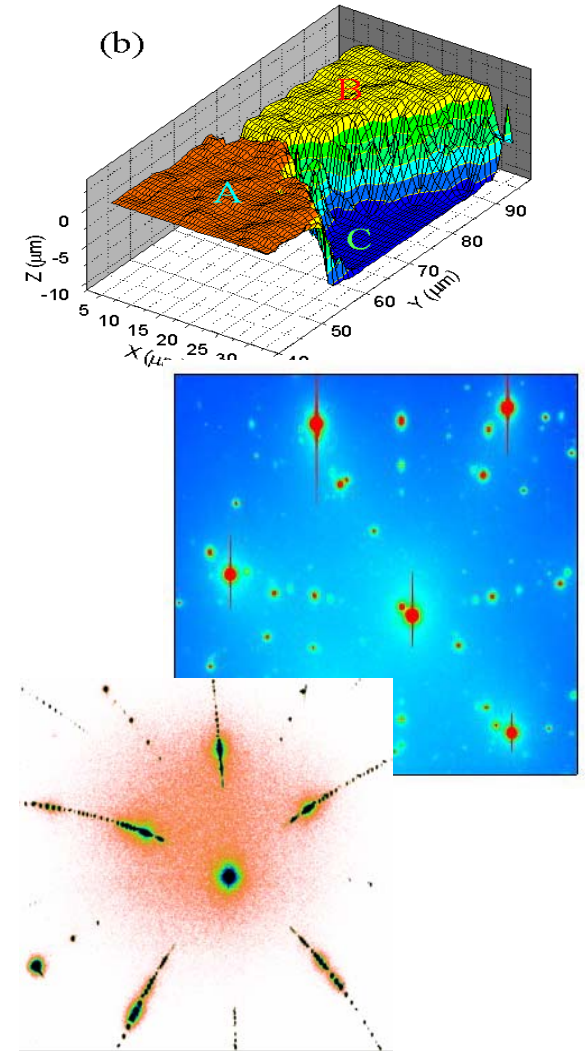
- Multiple wire/coded aperture
- Faster detectors (GE detectors)





# Summary: - important emerging technique

- Cannot meet demand with existing facilities
- Addresses long-standing issues with fundamentally new approach
- Wide applicability



# Team of ORNL scientists involved

- Gene Ice- **Co-principle investigator, x-ray optics**
- Bennett Larson- **Co-principle investigator-3D deformation/nanoindentation**
- John Budai-**Epitaxial films and 3D grain growth**
- Jonathan Tischler-**Mesoscale measurements and computer analysis (CMSD - APS Site)**
- Wenge Yang-**Mesoscale deformation using nanoindentation (Guest Scientist- APS Site)**
- Wenjun Liu-**Grain boundary networks (Post Doc- APS Site)**
- Judy Pang-**in-situ 3D polycrystalline deformation**

*Important support from APS-differentially deposited elliptical mirrors and beam stabilization*

